



Article

Machine learning in high-entropy alloys: phase formation predictions with artificial neural networks

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ABSTRACT

Due to their complex compositions, high entropy alloys (HEAs) offer a diverse range of material properties, making them highly adaptable for various applications, including those crucial for future sustainability. Phase engineering in HEAs presents a unique opportunity to tailor materials for environmentally friendly technologies and energy-efficient solutions. However, the challenge of predicting phase selection, a key aspect in harnessing the full potential of HEAs for sustainable applications, is compounded by the limited availability of HEA data. This study presents a distinctive approach by using a precisely produced and selected dataset to train an artificial neural network (ANN) model. This dataset, unlike prior studies, is uniquely constructed to contain an equal amount of training data for each phase in HEAs, which includes single-phase solid solutions (SS), amorphous (AM), and intermetallic compounds (IM). This methodology is relatively unexplored in the field and addresses the imbalanced data issue common in HEA research. To accurately assess the model's performance, rigorous cross-validation was employed to systematically adapt the model's hyperparameters for phase formation prediction. The assessment includes metrics such as phase-wise accuracy (AM 86.67% SS 81.25% & IM 82.35%), confusion matrix, and Micro-F1 score (0.83), all of which collectively demonstrate the effectiveness of this approach. The study highlights the importance of feature parameters in phase prediction for HEAs, shedding light on the factors influencing phase selection. Its balanced dataset and training method notably advance machine learning in HEA phase prediction, providing valuable insights for material design amidst challenges and data scarcity in the field.

1. Introduction

Recently, Multi-principle element alloys (MPEAs) have been different from conventional metal alloys, as these alloys consist of an equal proportion of individual principal elements [1]. MPEA is commonly mentioned interchangeably with high entropy alloy (HEA) in the literature [2–4]. Due to its remarkable properties, high entropy alloys are characterized as novel and promising materials class. These alloys tend to have complex chemical compositions containing several components [5–8]. Nonetheless, the HEA definition limits the number of species to a minimum of four. In comparison, only two species of identical atomic

concentrations can comprise an MPEA. We opt to constantly use the term HEA in this paper because of its broader classification [3, 9]. Phase engineering is a strategic approach that employs various phase structures found in HEAs to achieve remarkable performance configuration [10]. This approach offers an abundance of potential to modify HEAs for specific applications, producing materials that are precisely tuned to meet various technological requirements. HEAs can exhibit a wide range of desirable characteristics, including elevated strength for high load-bearing capacity, increased hardness for improved durability, heightened ductility for improved deformability, robust wear resistance against

abrasion, immense environmental corrosion resistance, and exceptional catalytic characteristics that enable various chemical reactions [11–23]. Specific mechanical properties can be targeted utilizing the phases present in HEAs. These phases consist of amorphous (AM), intermetallic compounds (IM), single-phase solid solutions (SS), and hybrid SS and IM phases [6,24–26]. Predicting phase selection is essential for designing HEA, though the mechanism behind predicting phase selection is crucial in tailored HEA design, yet the mechanisms underlying phase formation are uncertain. Additionally, the properties of HEAs are significantly impacted by the phase structure, and despite improvements, designing HEA phases is still challenging and time-consuming [1, 5, 6].

Machine learning has become an important tool to help with material design [27–29]. Machine learning, inclusive of deep learning, requires extracting features from large datasets to recapitulate the relationships, which also offers the chance to predict the phase formation of HEAs focusing on existing research data by using a range of deep learning. Several studies reveal intriguing outcomes in phase formation prediction by compiling data on HEAs and developing deep learning algorithms [30, 31]. For predicting phase formation, Zhu et al. [5] have introduced a deep neural network (DNN) architecture using a residual network (ResNet), which achieves 81.9% overall accuracy. An ANN model is utilized in Islam et al.'s study for phase prediction, with 99% accuracy on training data, while the practical prediction accuracy was below 80%. [1]. Several algorithms were employed, including logistic regression, decision tree, support vector machine (SVM) classifier, random forest, gradient boosting classifier, and ANN in Y.V. Krishna et al.'s research work [32], ANN has demonstrated the best accuracy of more than 80% for the test data among these algorithms. New alloys were synthesized and characterized to validate the predictions that ANN is the most accurate prediction method in the studied alloy system. K-nearest neighbors (KNN), SVM, and ANN are the three machine learning algorithms used in the study by Huang et al. [33], and ANN exhibits superior testing accuracy than other models for predicting phases in new HEAs. In Uttam et al.'s [31] study, the use of a neural network (NN) model is introduced for the first time to predict the hardness of a refractory high entropy alloy (RHEAs), and the prediction is verified through experimental synthesis and microstructural analysis. This model successfully applies to various alloys to predict hardness, which is consistent with available experimental results. In another recent study [24], the predictive accuracy of an ANN model in determining phase selection across three distinct alloy types is evaluated. It emphasizes the extensive impact of atomic size differences (δ) on the phases within AM, SS, and IM alloys. The research effectively forecasts the phases in two novel alloys by leveraging this learning model in conjunction with a combination of three or four key parameters with confirmation through X-ray diffraction. This approach provides a potentially promising tool for advancing the composition design and phase selection of novel alloys. While the field is extending, challenges continue in predicting HEA phases, such as advancing deep learning algorithms and dealing with a lack of experimental data. Given the vast unexplored compositional design space, developing efficient

machine learning algorithms based on existing data becomes crucial for precise HEA phase prediction.

This work presents an ANN model that tackles the challenge of training with balanced data for each phase in HEAs, contrasting the frequent problem of unbalanced datasets in past research. The objective is to determine the hyperparameters that maximize predictive accuracy and generality when predicting phase selection in new HEAs, utilizing a balanced dataset. In this study for HEA phase formation prediction, an ANN model architecture is optimized for the current balanced dataset and fine-tuned hyperparameters such as batch size, learning rate, epochs, and dropout rate. The model's performance is evaluated on the final test set by measuring phase-wise accuracy, originating a confusion matrix, and determining the Micro-F1 score, and the results are then compared with prior studies. The study also analyses the significance of feature parameters in phase prediction outcomes, clarifying the relative importance of physical parameters influencing phase selection.

2. Computational Methods

The HEA dataset underwent preprocessing and preparation using conventional data science techniques prior to being utilized to train the model. Three distinct datasets from various earlier studies [34–36] were selected and employed to construct the ANN-based model. A dataset of 240 HEAs was obtained after the elimination of redundant samples and sections with incomplete or duplicated data. An instance of randomly selected five rows of the dataset is presented in the Pandas DataFrame format in Table 1. In this dataset, there are an equal number of 80 data points for each of the AM, SS, and IM phases. The dataset consists of 240 instances and 6 features, including 1 categorical feature designating different phases (AM, SS, or IM) and 5 numeric features representing valence electron concentration (VEC), the difference in electronegativity difference ($\Delta\chi$), atomic size difference (δ), mixing enthalpy (ΔH_{mix}), and mixing entropy (ΔS_{mix}).

Table 1. A glimpse of Pandas displaying random 5 instances of the data employed in this study. The units for ΔH_{mix} and ΔS_{mix} are kJmol^{-1} and $\text{JK}^{-1}\text{mol}^{-1}$, respectively

Data serial	Material	VEC	$\Delta\chi$	δ	ΔH	ΔS
145	CuNbNiTiZr	6.800	0.222225	9.258017	-21.280000	13.380867
125	Fe75P16B6Al3	7.070	0.144184	10.139168	-25.004400	6.509662
132	Ni45Ti20Zr27Al8	6.620	0.246144	10.746126	-45.303600	10.282708
222	CoCrFeNiMo0.1	8.200	0.110000	1.960000	-3.900000	12.200000
64	AlCrFeMo0.8Ni	6.625	0.188403	6.015400	-11.145833	13.350675

The labels were encoded into integers and assigned the values 0 for AM, 1 for SS, and 2 for IM phases, respectively, to denote the alloy phases in the ANN model. The formulas provided below are used to compute the numeric values for the five features [37–40].

$$VEC = \sum_{i=1}^n c_i VEC_i \quad (1)$$

$$\Delta\chi = \sqrt{\sum_{i=1}^n c_i (\chi_i - \bar{\chi})^2} \quad (2)$$

$$\delta = 100 X \sqrt{\sum_{i=1}^n c_i (1 - r_i/\bar{r})^2} \quad (3)$$

$$\Delta H_{mix} = \sum_{i=1, i < j}^n 4H_{ij} c_i c_j \quad (4)$$

$$\Delta S_{mix} = -R \sum_{i=1}^n c_i \ln c_i \quad (5)$$

Here, c_i (where $0 < c_i < 1$) denotes the atomic concentrations of the i -th element, while n represents the total number of components within a HEA. VEC_i and r_i signifies the atomic concentration, VEC , and atomic radius of each species of the i -th element and R denotes the gas constant. Using Miedema's model, the enthalpy of atomic pairs, is calculated [41]. $\bar{\chi}$ and \bar{r} refer for the weighted Pauling electronegativity and atomic radius, respectively, written as follows.

$$\bar{\chi} = \sum_{i=1}^n c_i \chi_i \quad (6)$$

$$\bar{r} = \sum_{i=1}^n c_i r_i \quad (7)$$

The data undergo preprocessing for feature values before training the architecture. Using the Pandas library [42], these values are normalized and scaled them between 0 and 1, as shown below:

$$X_{new} = \frac{X_i - X_{min,i}}{X_{max,i} - X_{min,i}} \quad (8)$$

Where, X_{new} represents to the normalized feature, X_i refers to the actual feature information, $X_{min,i}$ and $X_{max,i}$ stand for the minimum and maximum values respectively. Through this normalization procedure, dimensionless numeric features are generated, which ensures effective uniform numeric scaling and consistent treatment of all features.

A layer of neurons performs computational task with the ANN model. the output of each neuron within the hidden layer is denoted by a_j , as expressed in the following equation.

$$a_j = \sum_{i=1}^n x_i W_{ij} + b_j \quad (9)$$

Where b_j designates the bias coefficients and W_{ij} corresponds to the weights of each input parameter x_i . Google's TensorFlow [43] is a well-recognized library in this field and based on that, the machine learning neural network architecture is used. Figure 1 shows the architecture used in this study which encompasses backpropagation functions and several hidden layers. Eq. (9) is used to calculate the value of a_j for each neuron which is related with connection-specific weights. The activation function takes it as an input value. Five features of parameters are encompassed as input and the three neurons denoting different phases are included in the output layer.

The leaky Rectified linear unit (LReLU) activation function has been applied within the hidden layers. The Rectified Linear Unit (ReLU) [44], illustrated in Figure 2, is a common and popular activation function in Neural Networks (NNs). By compelling precise tuning of the learning rate, it can extend past predefined bounds during the NN training process because of its easiness and subsequent reduction in training computation time. Due to this issue, the activation function remains inactive for the neurons within the negative region during the training process. By assigning a small constant value, like 0.2, to the negative region, Leaky ReLU

(LReLU) [45], solves this, as presented in Figure 2. Three nodes of the model's output layer represent the alloy phases which receive input from the final hidden layer and then these nodes employ activation functions to predict the phase. The broadly used activation function, SoftMax, illustrated in Figure 3, was utilized in the output layer for this classification. The probability of the input belonging to different classes is illustrated by this normalized exponential function illustrates. Generally, the SoftMax function [46] is expressed as:

$$\sigma(y'_i) = \frac{e^{y'_i}}{\sum_{i=1}^n e^{y'_i}} \quad (10)$$

Here, $\sigma(y'_i)$ denotes the subsequent probability and the prediction vector is referred as y'_i . Consequently, the model's output was contrasted with the target labels to assess network's error. Here, cross-entropy [47] is utilized as cost (loss) function, which resembles to the negative logarithm of probability, and the following equation represents the function.

$$H_y(y') = - \sum_{i=1}^n y \log(\sigma(y'_i))$$

Here, y' refer to the prediction and y stand for one of the three target vectors. The neural network's final output is converted into a probability, and then, using cross-entropy, it is utilized to calculate the loss. The deviation between the actual distribution and the model's expected output distribution is computed by Cross-entropy. Afterward, the gradient descent algorithm is used, utilizing a learning rate of 0.013, to convey back this error through the network. The weights and bias are initiated randomly in the beginning of the training process. The loss function is minimized by adjusting them at each epoch. The accuracy of the network is quantified by the number of successful determinations of the target. The hyperparameter configuration for the artificial neural network (ANN) model encompassed a range of values and architectures, contributing to the systematic tuning process. For the number of hidden layers, the model was experimented with settings ranging from 3 to 5 layers, exploring the impact of network depth. Regularization techniques, such as L1 and L2, were introduced within the ranges of 0.01 to 0.025, allowing for the assessment of their influence on model generalization. Similarly, the learning rate, an essential factor in optimization, was varied between 0.001 and 0.013 to identify the optimal balance between convergence and avoiding local minima. Dropout rates, a regularization method to mitigate overfitting, were adjusted within the range of 0.1 to 0.4. Different batch sizes, ranging from 8 to 120, were examined to evaluate their effect on model training efficiency and convergence. These diverse configurations and their corresponding results constituted a comprehensive exploration of the ANN's hyperparameters to achieve the best predictive performance and phase wise accuracy. After training the model with the training dataset, a distinct, unseen test set, which was preserved during training, was used to evaluate the model. The feedback from validation provides guidance to adjust the parameter. The best model is chosen after demonstration of the optimum validation result. The hyperparameters are described in Table 2 and the test set is used to evaluate this model.

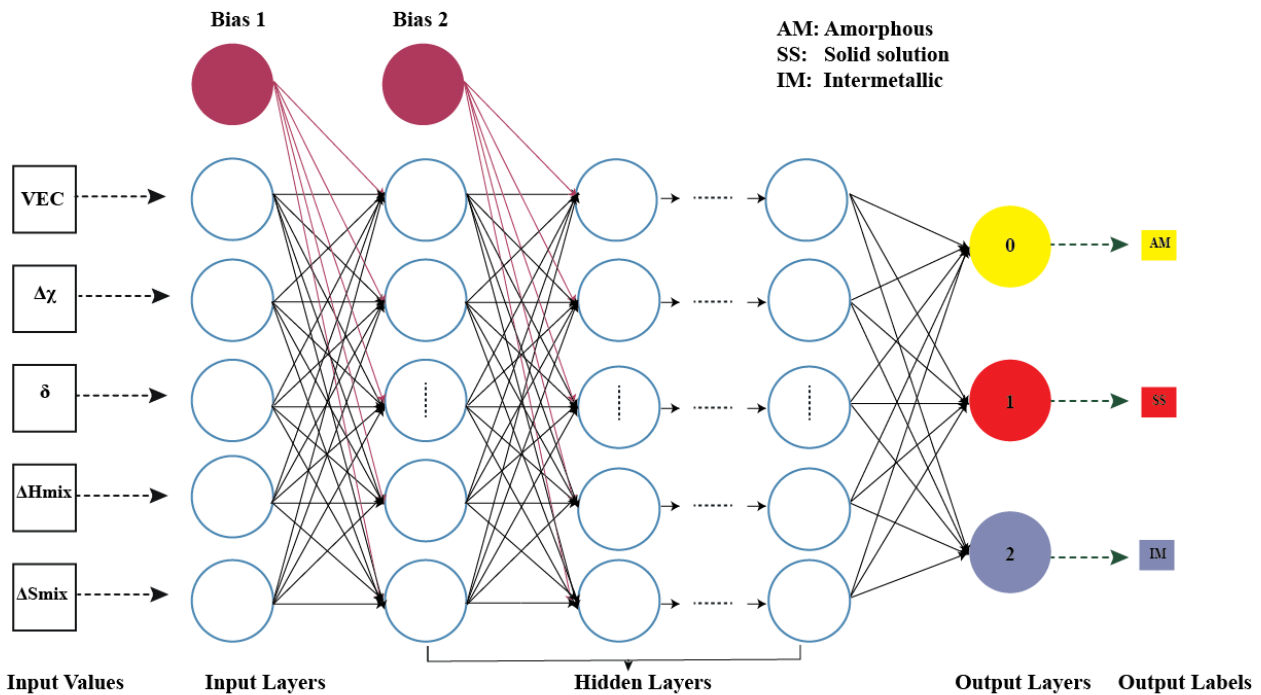


Figure 1. Illustration of the artificial neural network (ANN) architecture designed for predicting phase formations in High-Entropy Alloys (HEAs). For clarity, only five neurons (illustrated as circles) within the hidden layers are depicted. Empty squares symbolize input features and output values. The AM, SS, and IM phases are encoded as vectors 0, 1, and 2, respectively.

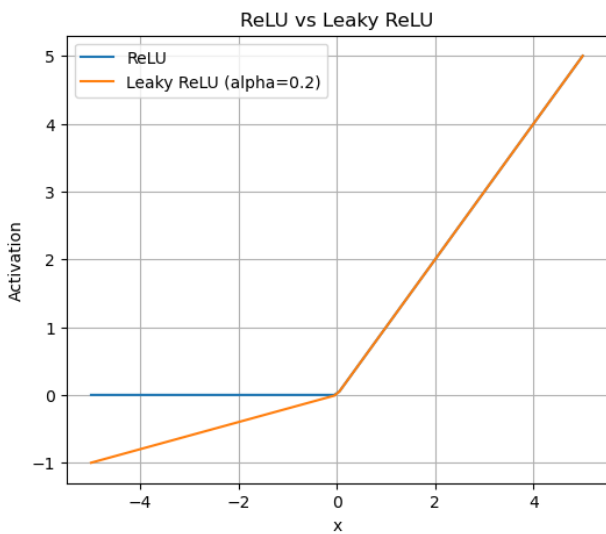


Figure 2. Rectified linear unit (ReLU) and leaky ReLU (LReLU)

3. Results and discussion

3.1 Data analysis

Comprehending the dataset of 240 records is an essential preliminary stage before applying a machine learning algorithm. Then, we generate a scatter matrix plot using the Seaborn package and then compute the correlation matrix of the features using Pandas library. These two matrices aid in comprehending feature relationships within the curated HEA dataset and offer both qualitative and quantitative

interconnection estimates. Our prediction pertains to the phases, with a specific emphasis on five quantitative features of the HEA compositions. To visualize the data, we employ a 5×5 scatter matrix plot, as depicted in Figure 4. The diagonal subfigures illustrate histograms of phase distributions, considering individual utilization of each of the five features. All histograms within subfigures overlap, suggesting no isolated feature for complete alloy phase classification. Correlations among the five features influence phase selection in HEAs, which is evident in off-diagonal subfigures of Figure 4.

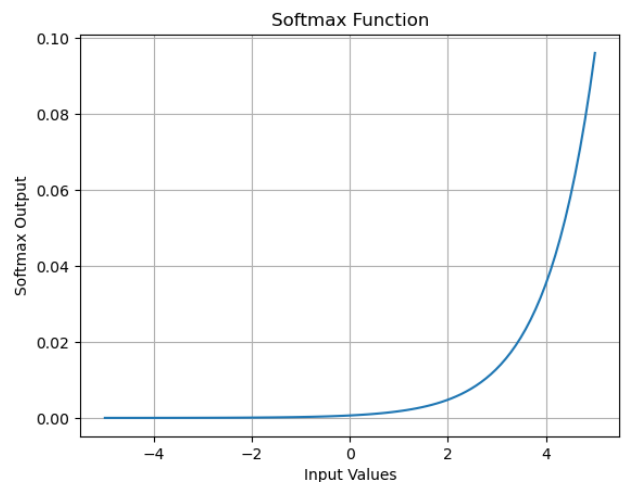


Figure 3. SoftMax function

Table 2. Hyperparameters of ANN

Hyperparameter	Value
Number of hidden layers	5
Number of hidden neurons	150 neurons
Regularization	L1: 0.025, L2: 0.01
Activation function	LeakyReLU (alpha= 0.1)
Dropout rate	0.4
Batch size	65
Learning rate	0.013 (Adam optimizer)
Epochs	100
Loss function	Categorical Crossentropy

We also calculate the Pearson correlation coefficient for features x and y to provide a quantitative description of their correlations [48].

$$r_{xy} = \frac{1}{n-1} \frac{\sum_{i=1}^n (x_i - \bar{x})(y_i - \bar{y})}{s_x s_y} \quad (12)$$

Here, x and y represent the mean values of two features, while S_x and S_y are their respective standard deviations. Correlation values can vary between -1 and 1, indicating negative or positive relationships. The computed correlation matrix elements are presented in Figure 5. Centering on the correlation between two distinct features, the matrix elements range from -0.61 to 0.72. Out of the ten distinct correlation matrix elements, seven exhibit negativity, while the remaining are positive. This outcome also exhibits resemblance to a prior study [1]. In the correlation matrix and scatter plot, of electronegativity difference ($\Delta\chi$) and atomic size difference (δ), a positive correlation is observed, meaning that $\Delta\chi$ tends to increase with higher values of δ . Additionally, both $\Delta\chi$ and δ show negative correlations with valence electron concentration (VEC) and mixing enthalpy (ΔH_{mix}). In general, the correlation matrix elements exhibit moderate magnitudes, allowing all five features to be employed collectively as input for our neural network architecture.

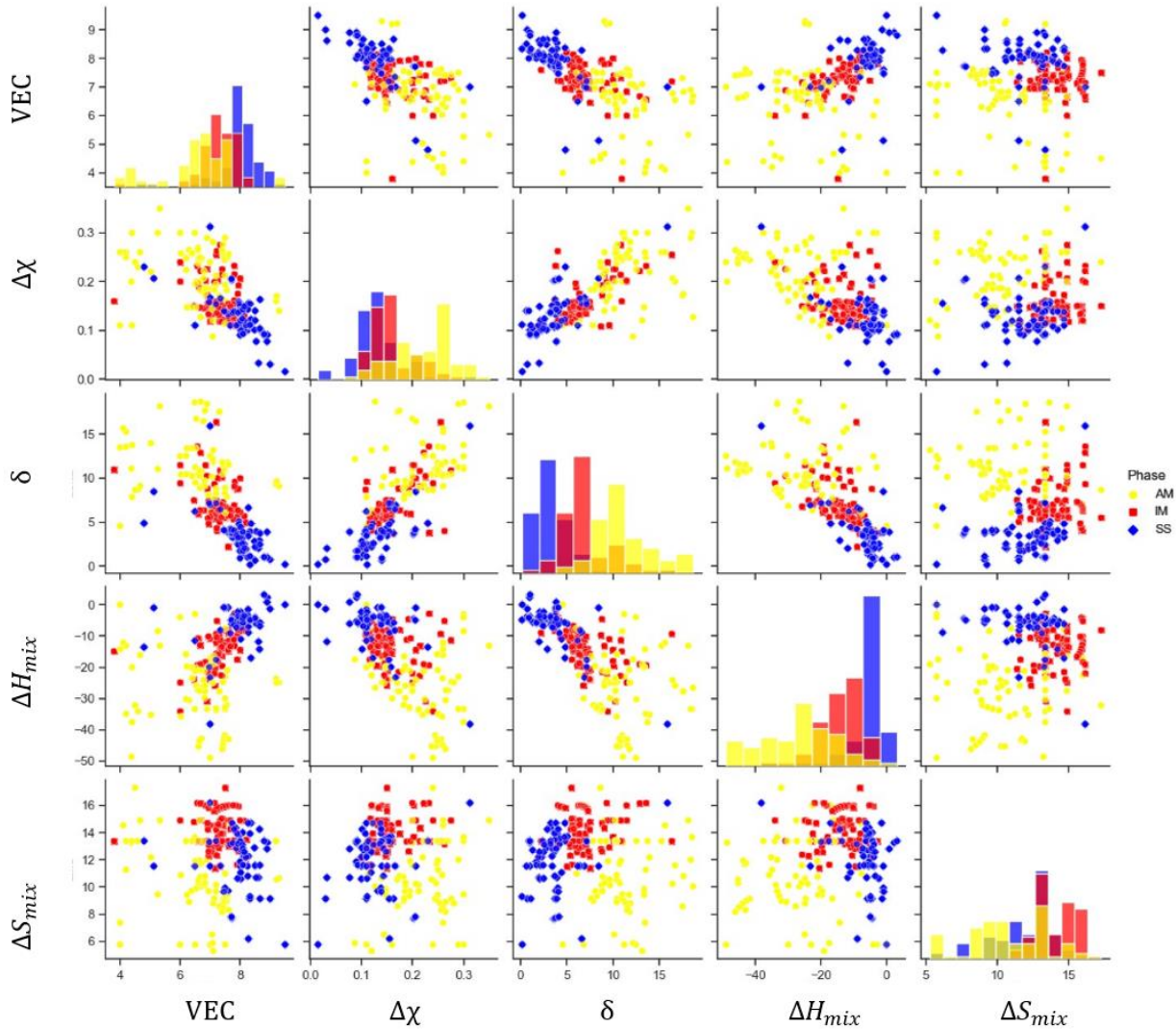


Figure 4. The scatter plots presented in the off-diagonal sections reveal the correlations among the values of the five distinct features. Within the diagonal panels, the histograms illustrate the distributions of the three phases based on the five features. Each phase is represented using varying shapes and colors: a yellow circle signifies amorphous (AM), a blue diamond represents solid solution (SS), and a red square denotes intermetallic (IM).

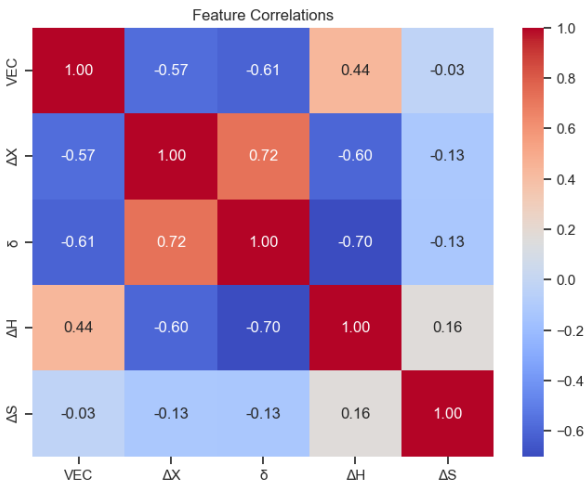


Figure 5. Correlation matrix heatmap of the five features

3.2 Artificial Neural Network (ANN) Results

The development of all Python network models is accomplished using the Keras framework with a TensorFlow backend for ANNs. Subsequently, the hyperparameter values are adjusted and the optimal model and hyperparameter settings for the ANN model are determined through a 3-fold cross-validation process as depicted in Figure 6. The best parameter resulted in an average cross-validation accuracy of 86.46%, as detailed in Table 2.

Afterwards, the ANN architecture is employed to train on 80% of the developed balanced dataset and then applied to test the remaining 20% of the dataset. This process is visualized in Figure 7(a), (b) and (c) illustrating the progression of training loss and validation loss across number of epochs for three separate folds.

Notably, both the training loss and validation loss curves exhibit a similar trajectory, demonstrating that the optimization algorithm consistently updates the weights of hidden layer neurons to minimize the loss and enhance the learning process at each epoch without overfitting the data. Across the three-fold training set, the loss converges to 1 after 40 epochs, maintaining a consistent trend thereafter. Furthermore, it's important to highlight that when the model is evaluated on the final set of data, there is a noticeable decrease in the loss value. This reduction brings the loss down to 0.5 after approximately 30 epochs of the training process. This trend is visually represented in Figure 7(d), where the curve illustrating the loss for the final test set which closely resembles the trajectory observed during the training process. This indicates that the model's performance on the test data doesn't show any signs of overfitting. This consistent reduction in loss and the convergence of the curves emphasize the model's ability to generalize well and perform effectively on new, unseen data for each of the HEA phases. In Figure 8(a), a visual representation is provided for the three-fold cross-validation process that was employed to assess the model's performance, showcasing the accuracy achieved for each individual fold along with average accuracy while training the data. Notably, the calculated average validation accuracy across all three folds was determined to be 86.46%. This approach of three-fold cross-validation ensures proper evaluation of the model's effectiveness across different subsets of the data. Additionally, the prediction of phase-wise accuracy on the training datasets is depicted in Figure 8(b). Remarkably, the final validation set attains a prediction rate nearing 83.33%, affirming the strong performance of the developed ANN model and its favorable generalization capabilities.

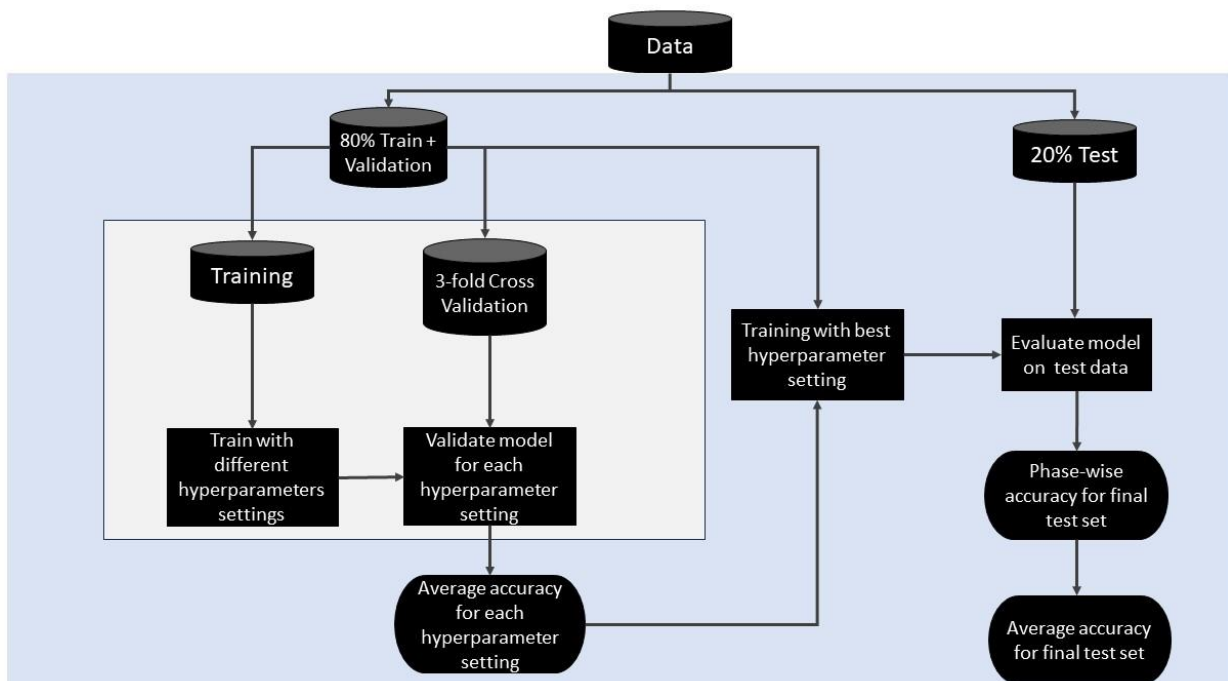


Figure 6. Training and testing process of the ANN model

Figure 9 illustrates the accuracy comparison between the training and validation processes for the final test set, revealing a positive correlation between them and overall accuracy improvement with epochs. Observing the training outcomes depicted in Figure 9, it's evident that the training set accuracy incrementally rises as iterations progress. After approximately 30 epochs, the accuracy stabilizes, suggesting effective convergence of the model. The effectiveness of the ANN model in predicting each of the phases is displayed in Figure 10. In contrast to some other studies that typically report overall accuracy, it is equally crucial to highlight phase-wise accuracy to illustrate the model's competence and its ability to predict various phases effectively. The results underscore the model's proficiency in accurately predicting distinct phases, and notably, the phase-wise accuracy reaches impressive levels, with AM achieving 86.67%, SS reaching 81.25%, and IM attaining 82.35%. These results exemplify the ANN model's effectiveness in predictive performance, particularly when it is trained on a balanced dataset for each phase. Using Micro-F1 to evaluate the prediction outcomes, the test set is employed to validate the effectiveness of the ANN model. The necessary equations for calculating the Micro-F1 score are provided below [5,49]: Using Micro-F1 to evaluate the prediction outcomes, the test set is employed to validate the effectiveness of the ANN model.

The necessary equations for calculating the Micro-F1 score are provided below [5,49]:

$$Recall_{micro} = \frac{\sum_{i=1}^n TP_i}{\sum_{i=1}^n TP_i + \sum_{i=1}^n FN_i} \tag{13}$$

$$Precision_{micro} = \frac{\sum_{i=1}^n TP_i}{\sum_{i=1}^n TP_i + \sum_{i=1}^n FP_i} \tag{14}$$

$$F1_{micro} = 2 \frac{Precision_{micro} * Recall_{micro}}{Precision_{micro} + Recall_{micro}} \tag{15}$$

Here, true positive (TP_i) means positive cases correctly identified as positive cases of the i element, false positive (FP_i) means negative cases is incorrectly identified positive cases of the i element, true positive (FN_i) means positive cases is incorrectly identified negative cases of the i element. $Recall_{micro}$ measures the accuracy of correctly predicting actual positive samples within the sample space. $Precision_{micro}$ quantifies the accuracy of forecasting positive predictions. $F1_{micro}$ is the aggregated average that considers both $Precision_{micro}$ and $Recall_{micro}$. The Micro F1 Score on final test set is 0.83. To assess the predictive performance of the ANN model for each of the HEA classes within the dataset, a confusion matrix was generated using a testing dataset comprising 48 samples, as illustrated in Figure 11.

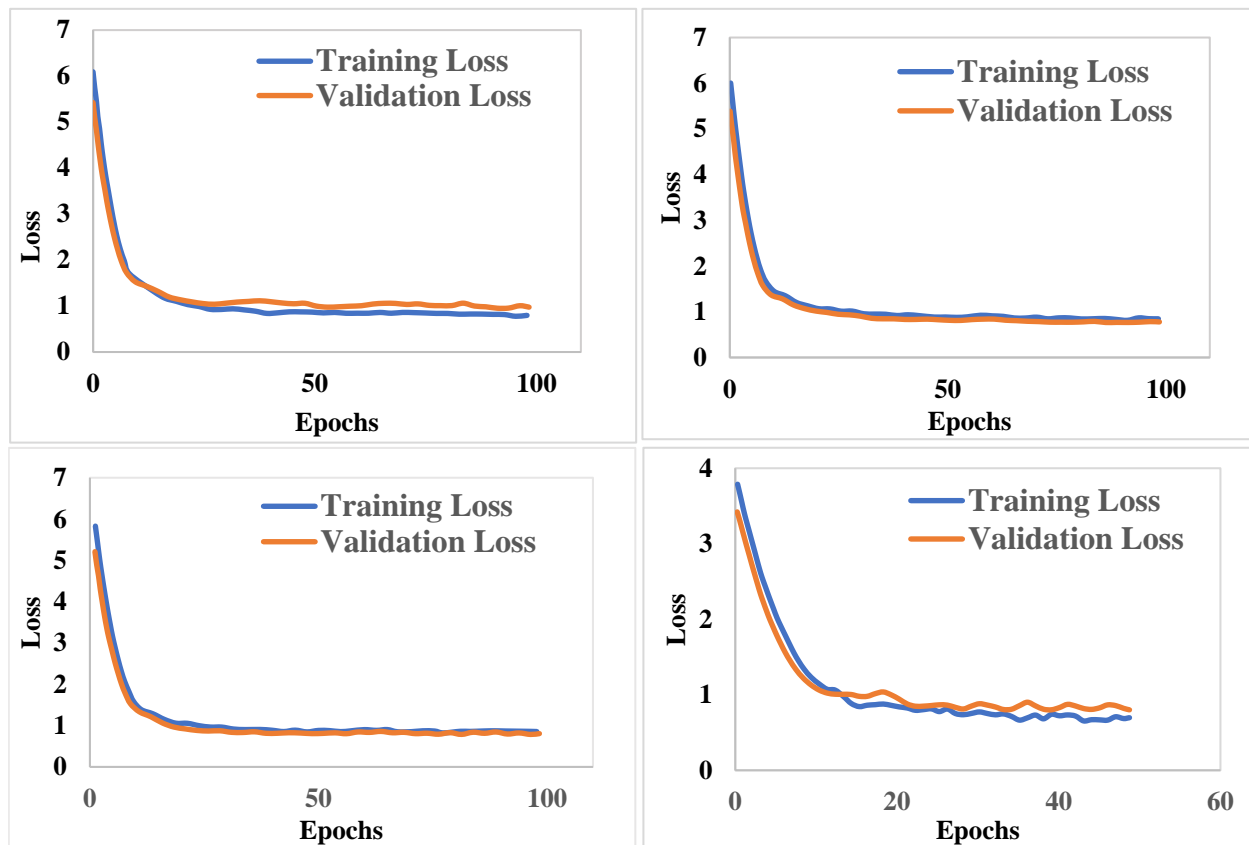


Figure 7. Comparing the training and validation loss of the ANN model for a) Fold 1, b) Fold 2, c) Fold 3, and d) Final test set.

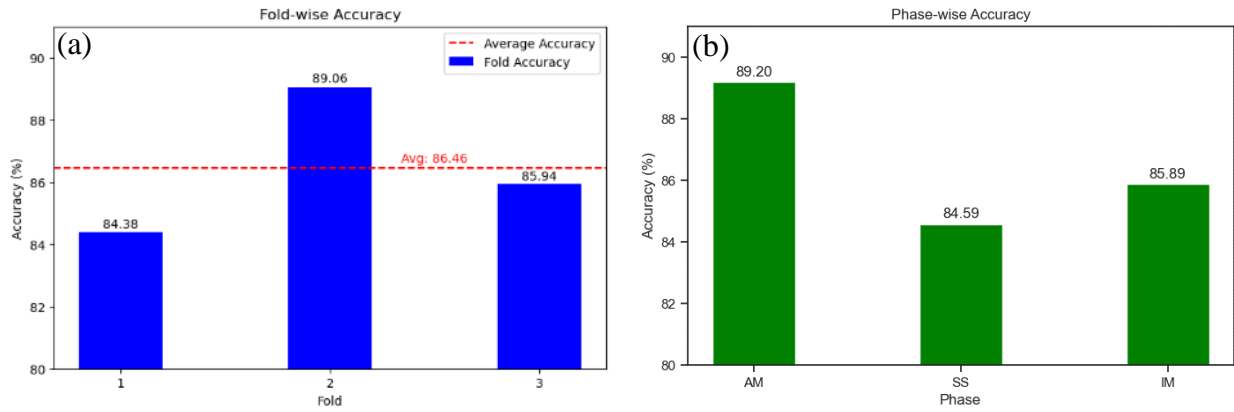


Figure 8. Comparing (a) fold-wise accuracy and (b) average phase-wise accuracy found for the cross-validation data employing the ANN model.

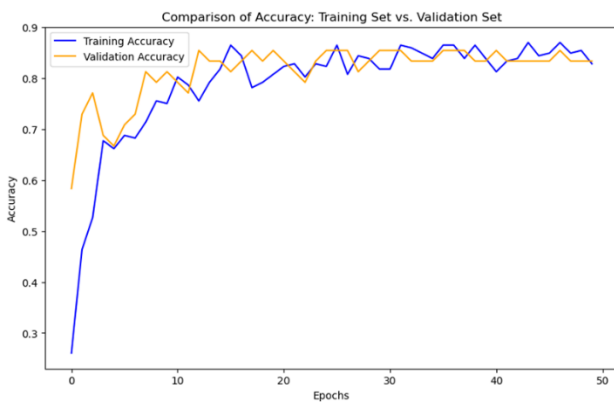


Figure 9. Comparing the accuracy of the ANN model between the training set and final validation set

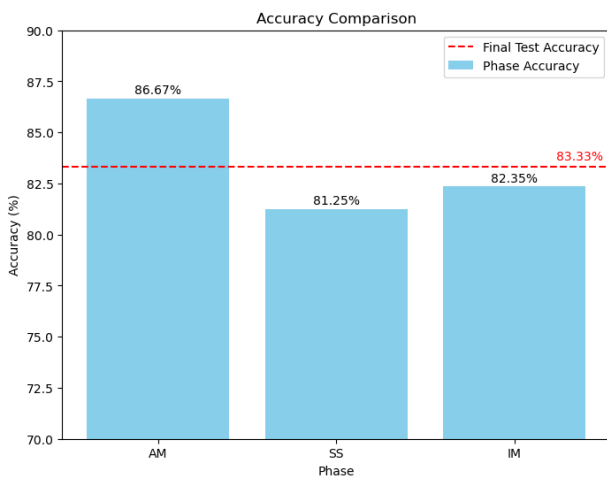


Figure 10. Comparing phase-wise accuracy and average accuracy found for the final test set

The confusion matrix indicates notably high precision and recall values across all three classes of HEA, affirming the model's robustness and generalization. The variance between the cross-validation accuracy and the confusion matrix accuracy arises because the former represents the average of all validation accuracies, while the latter directly reports

accuracy for the testing dataset that has been kept separate during the model development process. The developed ANN model aims to predict the phase of previously unseen data, and its performance was benchmarked against other alternative methods, as illustrated in Figure 12. Among the various machine learning algorithms assessed for accuracy, our developed ANN model demonstrates the highest accuracy of 83.33%. Notably, while Islam et al. [1] and Krishna et al. [32] also utilized an ANN model, their datasets exhibit unequal proportions of data across different HEA phases. In contrast, our ANN model maintains consistency by employing the same number of instances for AM, IM, and SS phases, and this uniformity contributes to the model's robust performance, allowing it to achieve the noteworthy accuracy of 83.33%, surpassing the accuracy of other methods [5,33,50], including those that employed differing dataset compositions. The developed ANN model aims to predict the phase of previously unseen data, and its performance was benchmarked against other alternative methods, as illustrated in Figure 12. Among the various machine learning algorithms assessed for accuracy, our developed ANN model demonstrates the highest accuracy of 83.33%.

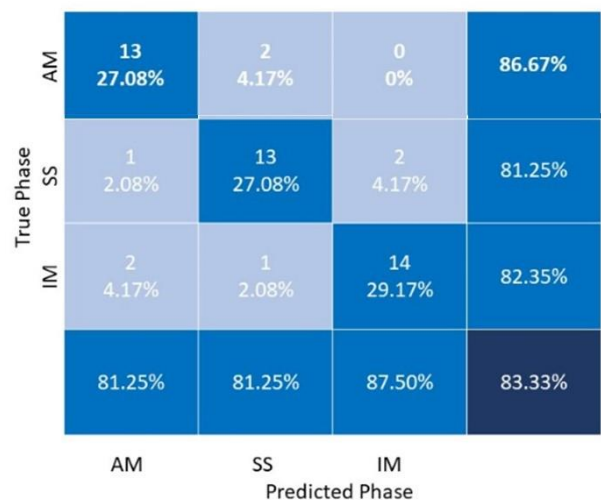


Figure 11. Confusion matrices of ANN model used in amorphous, solid solution, and intermetallic phase prediction on Final Test Set.

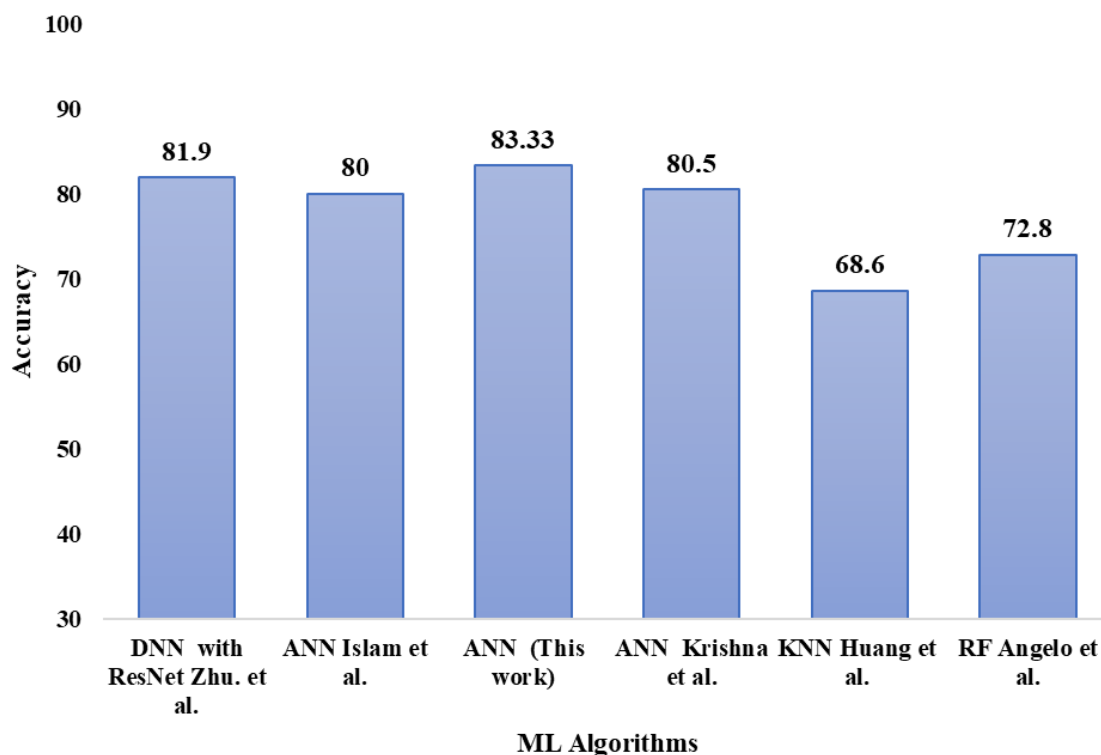


Figure 12. Evaluating the Precision of Machine Learning Algorithms

Notably, while Islam et al. [1] and Krishna et al. [32] also utilized an ANN model, their datasets exhibit unequal proportions of data across different HEA phases. In contrast, our ANN model maintains consistency by employing the same number of instances for AM, IM, and SS phases, and this uniformity contributes to the model's robust performance, allowing it to achieve the noteworthy accuracy of 83.33%, surpassing the accuracy of other methods [5,33,50], including those that employed differing dataset compositions.

3.3 Relative feature impact assessment

The ANN architecture was used to assess the relative significance of the five input features used to train the model. To investigate this, a series of five experiments were carried out, with each experiment systematically omitting one feature while keeping the remaining four. This procedure entailed retraining the model and making predictions in order to thoroughly investigate the impact on the test set accuracy. The results of these experiments can be seen in Figure 13, which depicts the decline in accuracy across the five scenarios mentioned above. This trend highlights an important observation: removing any of the five features consistently resulted in a decrease in the accuracy of the model, highlighting the significant influence that each feature has on test accuracy [5,25]. When compared to other features, differences in atomic sizes and the concentration of valence electrons have a greater influence on the accuracy of the model. Notably, it has been determined that the key design parameters derived from the current ANN approach—the atomic size difference and the valence electron concentration—align closely with the preexisting parametric guidelines for HEA phase formation.

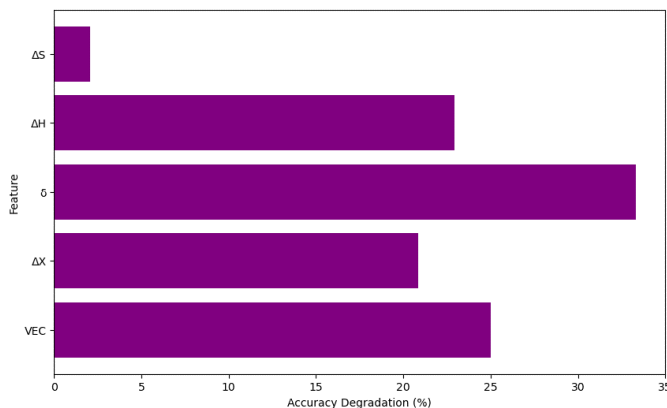


Figure 13. Effect on the test set accuracy upon the removal of individual features.

This convergence highlights an intriguing correlation, confirming the developed ANN method's reliability. Consistent with Hume-Rothery principles, the atomic size difference plays a crucial role in phase formation of HEA, especially in case of solid solution (SS) phase [5,38,51,52]. Furthermore, the Hume-Rothery principles show that the number of valence electrons per atom is critical in determining the stability of solid solutions in metal binary systems [5,52–54], and this stability in the mentioned systems hinges on electron density, specifically where peaks in the density of states occur, coinciding with the point where the Fermi sphere intersects the Brillouin zone boundary. As a result, the structure becomes stable at a specific electron concentration level. While atomic radius and

electronegativity differences are not always conclusive predictors of outcomes, they're both highly indicative parameters in the design of HEA compositions [51–54], underscoring the significance of considering electronic structure alongside other material properties when designing HEAs.

4. Conclusion

In this study, a carefully developed ANN model was introduced to address the persistent challenge of imbalanced datasets when predicting phase selection in HEAs. Through a rigorous optimization process encompassing various hyperparameters, the ANN model was developed using a balanced dataset, resulting in excellent predictive performance. Using a three-fold cross-validation strategy, the model's effectiveness was carefully evaluated. The results showed an impressive average validation accuracy of 86.46% across all three folds and eventually led to a high prediction rate of nearly 83.33% on the final test set, highlighting the model's robustness and capacity for generalization. This study also emphasized the importance of phase-wise accuracy, with the ANN model achieving remarkable accuracy levels for the studied HEA phases (86.67% for AM, 81.25% for SS, and 82.35% for IM). A detailed confusion matrix analysis also confirmed the model's robustness across all classes, highlighting its precision and recall balance, while comparison against alternative methods demonstrated its superior accuracy, and the Micro-F1 score validated the model's effectiveness with a score of 0.83 on the final test set. Notably, this was accomplished by maintaining dataset balance for each phase, which distinguished this approach from previous studies that frequently used imbalanced datasets and didn't mention phase-wise accuracy which is very important to showcase the model's ability for generalization. Furthermore, the study investigated the relative importance of input features, identifying that atomic size difference and valence electron concentration played critical roles in test accuracy, in line with established guidelines for HEA phase formation and reinforcing the developed ANN method's reliability. It should also be noted that including more data for each phase can contribute to even higher model performance, providing an exciting potential for further predictive accuracy improvement. Overall, this study not only provides an effective solution to an existing issue in materials science, but it also provides critical insights into the impact of physical parameters on phase selection, making it invaluable for future alloy design and engineering efforts.

Ethical issue

The authors are aware of and comply with best practices in publication ethics, specifically with regard to authorship (avoidance of guest authorship), dual submission, manipulation of figures, competing interests, and compliance with policies on research ethics. The authors adhere to publication requirements that the submitted work is original and has not been published elsewhere.

Data availability statement

The manuscript includes existing data, and additional data that support the findings of this study are openly available in High-Entropy-Alloy-Phase-Prediction-Using-Balanced-

Dataset at <https://github.com/fahel-bin-noor/High-Entropy-Alloy-Phase-Prediction-Using-Balanced-Dataset>.

Conflict of interest

The authors declare no potential conflict of interest.

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